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ABSTRACT

We present an efficient surrogate for binary-collision Monte-Carlo (BCMC) calculations in polyatomic materials for low ion energies based on Parkin and Coulter’s net displacement function. Most of BCMC’s execution time is spent on the low energy tail of cascades when recoils’ residual travel distance is small compared to the desired resolution of the damage calculation. We propose to curtail an ions’ random walk if its energy reduces below a threshold E_t and locally deposit an estimate of the expected number of defects it creates; the estimate of the number of defects is obtained from Parkin and Coulter’s net displacement functions. We find that results obtained with the BCMC code *MyTRIM* with and without truncation of the random walks compare well for large energy thresholds $E_t > 5$ keV. For $E_t = 5$ keV larger discrepancies are observed because of large differences between the net displacement function and *MyTRIM* for very low energy recoils. Future work will investigate and resolve this discrepancy. The energy threshold should be chosen so that ions’ average distance to next nuclear scattering is small compared to the desired resolution. A reduction of execution time of damage cascades by over 90 % is observed for an energy threshold of $E_t = 5$ keV and a 1 MeV Xe-135 primary impinging on UO_2 . The desired application of this surrogate model are coupled FEM-BCMC calculations with varying compositions where net displacement functions are tabulated by composition. For reducing the number of required net displacement functions, derivatives of the net displacement function with respect to number fraction are computed.

KEYWORDS: Radiation damage, Binary-collision Monte-Carlo, Computational Material Science, Net Displacement Function

1 INTRODUCTION

The number of displacements caused by primary knock-on atoms (PKA) is of central importance for predicting the evolution of material properties under irradiation. The binary-collision Monte-Carlo (BCMC) method [1] can be used to simulate representative radiation damage cascades caused by an atom of given type and initial energy. The *SRIM* code [1] allows computation of spatial distributions of point defects in one-dimensional geometries for polyatomic, amorphous

materials. The *MyTRIM* [2] code allows BCMC simulations in general, three-dimensional geometries. It is designed to be coupled to mesoscale simulations of nuclear fuel performed with Marmot [3, 4]. The coupling of Marmot and *MyTRIM* is accomplished by the Magpie (*Mesoscale Atomistic Glue Program for Integrated Execution*) code [5] that interfaces between the finite element method (FEM) description in Marmot and the discrete representation of defects in *MyTRIM*. In Magpie, *MyTRIM* computes point-defect creation rates that are rasterized on a FEM mesh, while Marmot updates material compositions that are used for computing ions' mean free paths.

Binary-collision Monte-Carlo simulations of damage cascades caused by fast ions are computationally expensive. A significant fraction of the execution time is spent on computing the motion of ions at low energies, i.e. late in their life cycle. The damage that these ions cause is spatially clustered and usually confined to a single FEM element. We propose to enhance the efficiency of BCMC calculations by truncating ion trajectories once their energy reduces below a threshold. The residual point defect production of these ions is estimated using a zero-dimensional radiation damage model.

For polyatomic materials, Parkin and Coulter provide integro-differential equations for the net displacement rate [6] defined as the number of atoms an ion of given type and energy displaces. The net displacement rate is a function of the primary ion's energy only, i.e. it is not a function of space, and the integro-differential equation contains both derivatives and integrals with respect to energy.

In this work, we first introduce the equation for the net displacement rate given by Parkin and Coulter [6]. We present an algorithm for numerically solving the equation for a given material composition. An online computation of the displacement rates is unfeasible and therefore we develop an efficient tabulation algorithm for displacement functions. Instead of storing displacement functions by position, we store them by composition signature and reuse them if the composition of interest is close to any of the compositions in the existing library. The required number of stored damage functions is reduced by using a linear approximation in composition space. The partial derivatives of the displacement function with respect to concentration are computed using linear perturbation theory.

2 NET DISPLACEMENT FUNCTIONS

The net displacement rate $g_{ij}(E)$ is defined as the number of atoms of type j displaced and not recaptured by PKA of type i and energy E [6]. The net displacement function $g_{ij}(E)$ satisfies the equation:

$$S_i(E) \frac{dg_{ij}}{dE} = \sum_{k=1}^K \frac{N_k}{N} [A_k(E_k^d, \Lambda_{ik}E) + B_k(0, \Gamma_{ik}(E)) - C_k(0, \Lambda_{ik}E)]$$

$$A_k(a, b) = \int_a^b \frac{d\sigma_{ik}(E, T)}{dT} g_{kj}(T - E_k^b) dT$$

$$B_k(a, b) = \int_a^b \frac{d\sigma_{ik}(E, T)}{dT} g_{ij}(E - T) dT$$

$$C_k(a, b) = g_{ij}(E) \int_a^b \frac{d\sigma_{ik}(E, T)}{dT} dT, \quad (1)$$

where $S_i(E) = N \left(\frac{dE}{dx} \right)_i$ is the electronic stopping cross section of species i moving with energy E , N_k is the number density of species k , N is the total number density of the background material, $\Lambda_{ik} = \frac{4A_i A_j}{(A_i + A_j)^2}$, A_i is the atomic mass of species i , $\frac{d\sigma_{ik}(E, T)}{dT}$ is the differential scattering cross section for interaction of species i and k (i : projectile, k target), E_k^d is the displacement threshold for species k , E_k^b is the binding energy defined as the energy that an atom of type k loses to inelastic processes and lattice vibrations as it is displaced from its lattice position. In addition, $\Gamma_{ik}(E)$ is defined by:

$$\Gamma_{ik}(E) = \min \left(\Lambda_{ik} E, \max \left(E_k^d, E - E_{ik}^{\text{cap}} \right) \right),$$

where E_{ik}^{cap} is the residual energy threshold of an atom of type i which has displaced an atom of type k to be trapped in the vacant k site. The initial conditions of Eq. 1 are given by $g_{ij} = \delta_{i,j}$ for $E < \min_i(E_i^d)$.

2.1 Electronic Stopping and Nuclear Scattering Cross Sections

Material properties enter the net displacement rate equation Eq. 1 in the form of the stopping power $S_i(E)$ and the differential nuclear scattering cross section $\frac{d\sigma_{i,k}(E, T)}{dT}$. Electronic stopping cross sections are taken from the *MyTRIM* code by dividing the stopping power by the total number density N .

The differential scattering cross section is computed using Lindhard's universal representation with the Thomas-Fermi potential [7]. The universal representation of the scattering cross section is given by:

$$\frac{d\sigma_{ij}(E, T)}{dT} = \frac{\pi a^2}{2T t^{1/2}} f(t^{1/2}), \quad (2)$$

where:

- $a = 0.8853 a_0 Z^{-1/3}$.
- $a_0 = 0.529177 \text{\AA}$: Bohr's radius.
- $Z = (Z_i^{2/3} + Z_j^{2/3})^{3/2}$.
- $t = \left(\frac{E}{E_L} \right)^2 \frac{T}{T_m}$.
- $E_L = \frac{Z_i Z_j e^2}{4\pi \epsilon_0 a} \frac{A_i + A_j}{A_j}$
- $\epsilon_0 = 8.85 \times 10^{-12} \frac{\text{F}}{\text{m}}$: vacuum permittivity.
- $T_m = \Lambda_{ij} E$.

and $f(\xi)$ changes depending on the interaction potential used to evaluate the cross section. For the Thomas-Fermi potential Winterborn, Sigmund, and Sanders [8] find that

$$f(\xi) = \gamma \xi^{1/3} \left[1 + (2\gamma \xi^{4/3})^{2/3} \right]^{-3/2}, \quad (3)$$

with $\gamma = 1.309$ approximates the exact $f(\xi)$ reasonably well. This work uses Eq. 3 throughout.

Numerical solution of Eq. 1 requires analysis of the asymptotic behavior of $\frac{d\sigma_{ij}(E,T)}{dT}$ as $E, T \rightarrow 0$. To this end, we evaluate Eq. 2 in the limit $t \rightarrow 0$:

$$\lim_{t \rightarrow 0} \frac{f(t^{1/2})}{t^{1/2}} = \gamma t^{-1/3}. \quad (4)$$

We find that for small E, T , Eq. 2 becomes:

$$\frac{d\sigma_{ij}(E, T)}{dT} = 1.309 \frac{\pi a^2}{2} \left(\frac{A_i}{A_j} \right)^{1/3} \left(\frac{2Z_i Z_j e^2}{4\pi \epsilon_0 a} \right)^{2/3} E^{-1/3} T^{-4/3}. \quad (5)$$

2.2 Numerical Solution of Net Displacement Equation

In this section the numerical issues with solving Eq. 1 are discussed. The main issue is that the interaction cross section becomes unbounded as $T \rightarrow 0$ but we show in this section that the right hand side of Eq. 1 remains bounded. We will start discussing the general numerical approach taken to integrate Eq. 1 first and then highlight the special treatment to ensure numerical stability.

Magpie uses algorithms implemented in the GNU scientific library (GSL) [9] for the solution of the relevant equations. Within this work, the explicit 4th order (classical) Runge-Kutta is used. Numerical integration of ODEs results in solution on an energy grid denoted by $E_n, n = 1, \dots, N$ where the solution at these points is denoted by $g_{i,j}^{(n)} = g_{i,j}(E_n)$.

Integrals over energy that appear on the right hand side of Eq. 1 are computed as sums of elementary integrals over the range E_n to E_{n+1} . The elementary integrals are estimated using GSL's Gauss-Legendre quadrature of order 4 and the value of $g_{i,j}(E)$ is interpolated linearly using $g_{i,j}^{(n)}$ and $g_{i,j}^{(n+1)}$.

Even though the scattering cross section becomes unbounded for $T \rightarrow 0$, the right hand side of Eq. 1 remains bounded. To demonstrate that, we first rearrange terms:

$$B_k(0, \Gamma_{ik}(E)) - C_k(0, \Lambda_{ik}E) = B_k(E_c, \Gamma_{ik}(E)) - C_k(E_c, \Lambda_{ik}E) + \int_0^{E_c} \frac{d\sigma_{ik}(E, T)}{dT} (g_{ij}(E - T) - g_{ij}(E)) dT, \quad (6)$$

where E_c is suitably chosen energy cutoff. The apparent singularity is now confined to the last term in Eq. 6. We will now show that Eq. 6 is bounded. Following [10] we expand the last term into a Taylor series expansion for small T :

$$g_{ij}(E - T) - g_{ij}(E) = -T \left. \frac{dg_{ij}}{dE} \right|_{T=0} + \mathcal{O}(T^2). \quad (7)$$

Given that the cross sections are of order $T^{-4/3}$, the integrand of the last term is of order $T^{-1/3}$ and thus the integral is bounded.

It is numerically most favorable to reformulate Eq. 1 by inserting Eq. 7 and solving for dg_{ij}/dE :

$$\frac{dg_{ij}}{dE} = \frac{\sum_{k=1}^K \frac{N_k}{N} [A_k(E_k^d, \Lambda_{ik}E) + B_k(E_c, \Gamma_{ik}(E)) - C_k(E_c, \Lambda_{ik}E)]}{S_i(E) + \sum_{k=1}^K \frac{N_k}{N} \int_0^{E_c} T \frac{d\sigma_{ik}(E,T)}{dT} dT}. \quad (8)$$

3 POLYATOMIC PARKIN-COULTER MODEL IN MAGPIE

The net displacement functions are used as a surrogate for detailed BCMC cascades below a selected energy threshold E_t . Once an ion's energy drops below the energy threshold, the random walk is terminated and an estimate of the expected number of vacancies and interstitials are deposited locally. The local vacancy and interstitial counts ν_j and η_j are incremented by:

$$\begin{aligned} \nu_j &:= \nu_j + g_{ij}(E) \\ \eta_j &:= \eta_j + g_{ij}(E), \end{aligned} \quad (9)$$

where i is the current ion's type and E its energy.

Displacement functions are tabulated by number fraction. If an ion's random walk is terminated, the local composition is compared with tabulated displacement functions. If a sufficiently good match exists, the existing net displacement function is used in Eq. 9, otherwise a new net displacement function is computed.

The described algorithm is tested on two systems, monoatomic metallic uranium and polyatomic UO_2 in a quasi one-dimensional geometry. For both systems, Xe-135 primaries with an energy of 1 MeV impinge on an infinitely thick target. We tally the linear density of vacancies as a function of penetration depth for different energy thresholds. For UO_2 , the total vacancy density is obtained by summing the uranium and oxygen vacancy densities.

The computed vacancy densities plotted versus penetration depth are presented in Figs. 1 and 2, respectively. Each plot contains the *MyTRIM* only calculation indicated by $E_t = 0$, *MyTRIM* calculations with random walks truncated at various $E_t > 0$, and finally a case $E_t = 5 \text{ keV}^*$ that stops ions at 5 keV but does not apply Eq. 9.

The vacancy densities with random walks truncated at E_t do not converge to the unmodified *MyTRIM* results as $E_t \rightarrow 0$. This is caused by large differences in the predicted number of defects produced by low energy ions. For 200 eV uranium projectiles, *MyTRIM* predicts an average of about 0.7 vacancies per projectile, while Eq. 1 predicts between 1.5 and 1.6 vacancies. The large discrepancy of the curves with $E_t = 5 \text{ keV}$ and $E_t = 25 \text{ keV}$ is explained as follows: during the damage cascade energy is distributed onto a growing number of particles, so that early in the cascade a few high energy ions exist, and late in the cascade many low energy ions exist. If the energy threshold is chosen at $E_t = 5 \text{ keV}$, we record a large number low energy contributions to ν_j , while we record a small number of large energy contributions at larger E_t . For small E_t the differences in *MyTRIM*'s and Eq. 1's vacancy yields are amplified by the large number of events leading to a poor prediction of vacancy densities. Currently, this discrepancy is under investigation. It is imperative for the deployment of the truncated cascades that we recover non-truncated *MyTRIM* results as $E_t \rightarrow 0$.

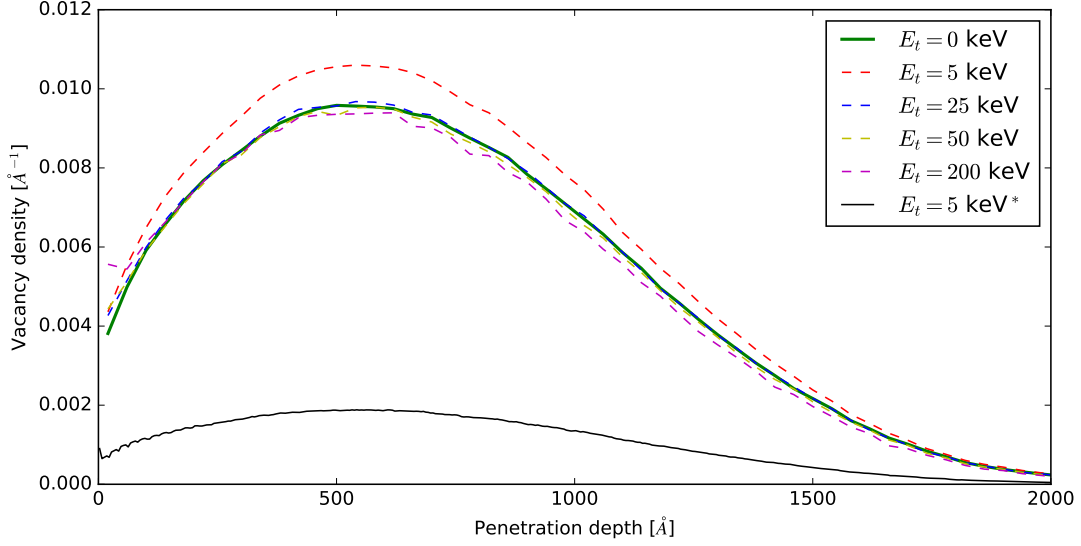


Figure 1: Linear vacancy density as a function of depth in metallic Uranium created by 1 MeV Xe-135 ions impinging on the left hand side of the domain. Random walks are terminated if the ion's energy drops below E_t and with the exception of $E_t = 5 \text{ keV}^*$ an estimate of future damage is deposited locally using Eq. 9.

Despite the poor prediction for small E_t , the polyatomic Parkin-Coulter formalism gives encouraging results for $E_t \approx 25 \text{ keV}$. The solid black curves in Figs. 1 and 2 are the vacancies produced by ions with energies larger than 5 keV. It is evident that most vacancies are created by ions with $E < 5 \text{ keV}$ and that truncation of their random walks without applying Eq. 9 leads to unacceptable errors.

The accuracy of the predicted spatial distribution of vacancies reduces as E_t is increased. Faster ions have larger mean free path. If a fast ion's random walk is curtailed and its future potential for vacancies deposited locally, then the spatial distribution of these vacancies is lost if the mean free path of the ion is comparable or larger than the size of the desired spatial resolution. The spatial resolution is set by the FEM element size that is fixed at 40 Å in this work. Uranium ions have a distance between two collisions of the order of $1 - 5 \text{ Å}$, while oxygen usually travels more than 10 Å between collisions. Therefore, we observe a trend to premature deposition of vacancies in the case of UO_2 , Fig. 2. In this case, the total number of predicted vacancies for $E_t = 0$ and $E_t = 25 \text{ keV}$ is very similar, but when truncating random walks vacancies are created at smaller penetration depths.

Truncating the random walk has the potential of saving orders of magnitude of execution time without sacrificing accuracy. A cascade of 1 MeV Xe-135 particles can be executed 12 times faster if $E_t = 5 \text{ keV}$. This is particularly important for coupled BCMC-FEM calculations where repeated damage cascades for many primary ions have to be completed.

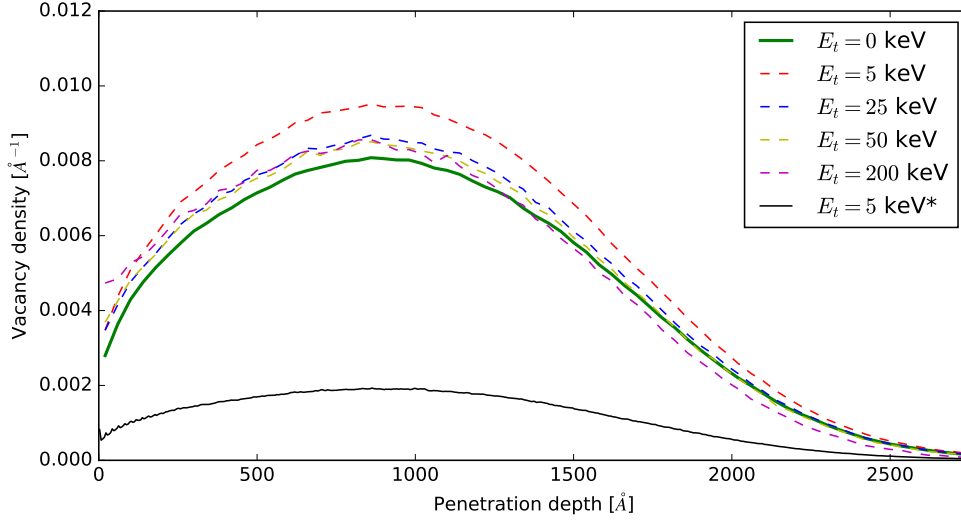


Figure 2: Linear vacancy density as a function of depth in UO_2 created by 1 MeV Xe-135 ions impinging on the left hand side of the domain. The vacancy density is the sum of oxygen and uranium vacancy densities. Random walks are terminated if the ion's energy drops below E_t and with the exception of $E_t = 5 \text{ keV}^*$ an estimate of future damage is deposited locally using Eq. 9.

4 FIRST ORDER PERTURBATION

In this section, we derive an equation for the derivative of the net displacement function with respect to the l -th number fraction, i.e. $\theta_{ijl} \equiv \frac{dg_{ij}}{df_l}$, where $f_l = N_l/N$. The purpose of computing the derivative is to estimate the changes of the net displacement function when the number fractions change. This approach allows to compute fewer net damage functions for spatially distributed radiation damage problems with continuously varying compositions. Future work will use the first order estimate as efficient surrogate for BCMC in coupled FEM-BCMC problems where the composition varies within the domain.

To first order, changing the number densities f_l to f'_l lead to the following changes in the net displacement rates:

$$g_{ij}(E, \vec{f}') = g_{ij}(E, \vec{f}) + \sum_{l=1}^K \theta_{ijl}(E, \vec{f})(f'_l - f_l) + \mathcal{O}((f'_l - f_l)^2). \quad (10)$$

The equations describing the derivative of the net displacement with respect to number fractions is derived using first order perturbation theory. We first introduce a small perturbation δf_l that causes a small perturbation in the net displacement function denoted by $\delta g_{ij}^{(l)}$. The perturbation of f_l is chosen to be small enough so that products of two perturbations can be neglected. Substituting:

$$\begin{aligned} f_k &\leftarrow f_k + \delta f_l \delta_{l,k} \\ g_{ij} &\leftarrow g_{ij} + \delta g_{ij}^{(l)} \end{aligned}$$

$$s_i \leftarrow s_i + \delta s_i^{(l)}, \quad (11)$$

into Eq. 1 leads to an equation for the ratio $\delta g_{ij}^{(l)} / \delta f_l$. The equation for θ_{ijl} is obtained by passing to the limit $\delta f_l \rightarrow 0$:

$$\begin{aligned} S_i(E) \frac{d\theta_{ijl}}{dE} &= \sum_{k=1}^K f_k \int_0^{\Lambda_{ik} E} dT \frac{d\sigma_{ik}}{dT} \left[\rho_k(T) \theta_{kjl}(T - E_k^b) \right. \\ &\quad \left. + (1 - \rho_k(T) \lambda_{ik}(E - T)) \theta_{ijl}(E - T) - \theta_{ijl}(E) \right] + Q_{i,j,l}(E) \\ Q_{i,j,l}(E) &= \int_0^{\Lambda_{il} E} dT \frac{d\sigma_{il}}{dT} \left[\rho_l(T) g_{lj}(T - E_l^b) + (1 - \rho_l(T) \lambda_{il}(E - T)) g_{ij}(E - T) - g_{ij}(E) \right] \\ &\quad - \frac{dS_i(E)}{df_l} \frac{dg_{ij}}{dE}, \end{aligned} \quad (12)$$

where:

$$\begin{aligned} \rho_k(T) &= \begin{cases} 0, & T < E_k^d \\ 1, & T \geq E_k^d \end{cases} \\ \lambda_{ik}(E) &= \begin{cases} 1, & T < E_{ik}^{\text{cap}} \\ 0, & T \geq E_{ik}^{\text{cap}} \end{cases}. \end{aligned} \quad (13)$$

The initial conditions for the net displacement rates are $g_{ij}(E) = \delta_{ij}$ for $E < \min_i(E_i^d)$ which are independent of f_l . Therefore, the initial condition for θ_{ijl} is $\theta_{ijl}(E) = 0$ for $E < \min_i(E_i^d)$. Equation 12 is identical in form to the original equation for the net displacement function, Eq. 1, except for the source term $Q_{i,j,l}$. Therefore, solution is straight forward by first obtaining g_{ij} and then using the same algorithm for computing θ_{ijl} with the only difference that Q_{ijl} is computed from the computed net displacement rate and then added to the right hand side.

We test the correct implementation using Al_2O_3 as an example. A base case calculation is performed using number fractions of 0.4 and 0.6 for Al and O, respectively; then the number fractions are modified to 0.3 and 0.7, respectively, corresponding to $\delta f_{\text{Al}} = -0.1$ and $\delta f_{\text{O}} = 0.1$. The net displacement rates are then (1) re-computed by Eq. 1 using updated number fraction and (2) estimated by Eq. 10. The results are presented in Fig. 3. The markers indicate the direct computation of g_{ij} with modified number fractions, while the dashed-dotted line indicates the estimate using first order perturbation (fop) theory. We observe that estimated and directly computed values agree very well. The limit to which we can use Eq. 10 in lieu of computing a new displacement function will be investigated for real systems in future work. It is expected to depend on the desired level of accuracy and the particular material system.

5 CONCLUSIONS

We present an efficient surrogate for low energy damage effects in polyatomic materials. A large portion of the execution time of BCMC calculations is spent on low energy ions that deposit their

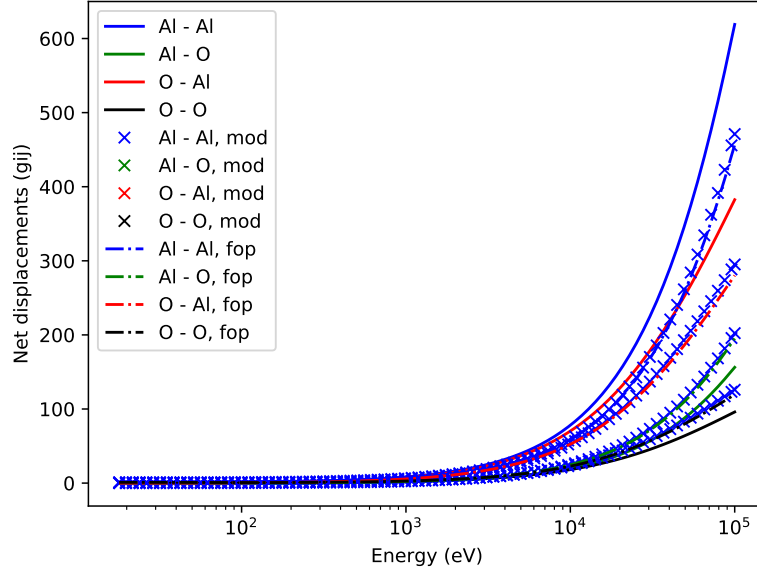


Figure 3: Testing the correctness and precision of estimating the change in net displacement rates of Al_2O_3 due to changes in number fractions using first order perturbation theory. The solid lines represent g_{ij} for the base number fractions 0.4/0.6, the markers indicate direct computation of the modified system with number fractions 0.3/0.7 (mod), the dashed-dotted lines indicate the estimation using first order perturbation theory (fop).

energy in close proximity of their current location. The Magpie application is designed for solving coupled BCMC-FEM problems, e.g. micro-structure evolution under irradiation. Low energy ions are likely to deposit their energy within a single mesh element so that detailed simulation of ions below a certain energy threshold does not add fidelity to the coupled calculation. Instead of following low energy ions, we propose to terminate ions' random walks and deposit an estimate of the remaining damage locally; the remaining damage is estimated using Parkin and Coulter's net displacement function. Damage functions are stored by composition and reused if an element's composition matches an existing entry's composition sufficiently well. We find that results obtained with the BCMC code *MyTRIM* with and without truncation of the random walks compare well for large energy thresholds $E_t > 5$ keV. For $E_t = 5$ keV larger discrepancies are observed because of large differences between the net displacement function and *MyTRIM* for very low energy recoils. Future work will investigate and resolve this discrepancy. The energy threshold should be chosen so that ions' average distance to next nuclear scattering is small compared to the desired resolution. A reduction of execution time of damage cascades by over 90 % is observed for an energy threshold of $E_t = 5$ keV for a 1 MeV Xe-135 primary impinging on UO_2 . A linear approximation for computing displacement functions in terms of the number fractions is derived to reduce the number of stored net displacement functions and will be used for coupled FEM-BCMC problems with varying composition.

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